

This result can now be inserted into the expression for M :

$$\begin{aligned} \lim_{b \rightarrow \infty} M_n^\gamma(a, b) &= \int_0^{2\pi} \exp [ia \cos \alpha] \lim_{b \rightarrow \infty} P(\alpha) d\alpha, \\ M_n^\gamma(a, \infty) &= (1/n) \int_0^{2\pi} \exp [ia \cos \alpha] \sum_{m=1}^n \delta(\alpha - \gamma - 2\pi m/n) d\alpha \\ &= (1/n) \sum_{m=1}^n \exp [ia \cos (\gamma + 2\pi m/n)], \end{aligned}$$

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Vector maps of hydrated protein crystals.* By DOROTHY WRINCH, *Department of Physics, Smith College, Northampton, Mass., U.S.A.*

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The difficulty of attempting structure analyses of protein crystals by studying the intensities recorded in X-ray diffraction studies has focussed attention on their vector maps. This note is concerned with the significance, for these vector maps, of the considerable complements of water in protein crystals.

We picture a crystalline protein as an array of molecules in a medium in atomic space S . To a first approximation, the medium may be represented by a constant m and the electron-density function $g(x, y, z)$ will take the value m over the intermolecular regions in the cell. Let us suppose that the same molecules crystallize in the same medium to form a series of hydrates, each with its own unit cell. Then a separate g is required for each hydrate and the question arises as to a method whereby the problem of calculating the functions $g_v(x, y, z)$ defining the vector maps can be treated as a whole. A schematic representation of such a situation is seen in Fig. 1 in which a and c are 'crystals' in which the same 'molecule' crystallizes in the same 'medium' ($m=1$); in A and C the corresponding maps in vector space S_v are separately calculated. It is evident that the problem is not solved simply by using the finite vector map of the common molecule, which is shown between A and C . To it we should need to add the vector map of the medium and the interactions of the medium and the molecule; these maps necessarily differ from crystal to crystal. However, progress can be made if we focus attention not on the common molecule but on the common medium. Accordingly we record, in b and d , the crystals reduced to the level of the medium, defined by the reduced density functions $r_m(x, y, z) = g(x, y, z) - m$. Each crystal is now represented simply by the original molecule reduced to the level m , as shown between b and d . For this reduced molecule, the vector map is recorded, once for all. Inserting this map in the appropriate orientation at given sets of lattice points, the vector maps of the reduced crystals are obtained as shown in B and D . To see the relation between the vector map of a reduced crystal and the vector map of the original crystal, we recall the theorem (Wrinch, 1939) that the vector maps of a crystal defined by $g(x, y, z)$ and by $g(x, y, z) - k$ are represented by

which is the desired result. The δ -function argument may, of course, be replaced by well-known limiting processes.

References

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$g_v(x, y, z)$ and by $g_v(x, y, z) - k_v$, where k_v is such as to make the total weight in the cell in S_v equal to the square of the total weight in the cell in S in each case. If there are n entries in the cell, we may write w and $w - nk$ for the total weight in S and $w_v (= w^2)$ and $w_v - nk_v$ for the total weights in S_v . It then follows that

$$w_v - nk_v = (w - nk)^2, \quad k_v = k(2w - nk).$$

In the figure we see an example of this theorem by comparing B with A and D with C .

It is thus possible to formulate the vector maps for crystals in which molecules are arrayed in a constant medium by studying the crystals reduced to the level of the medium, and it is sufficient to record simply the vector maps of the reduced crystals since each is the vector map of the original crystal reduced to a certain level. The examples in Fig. 1 confine attention to the case of one molecule in the unit cell, and the only preliminary to writing down the vector maps is the vector map of the molecule reduced to the m level. This procedure may be extended to cover any number of molecules. When the crystal reduced to the level m comprises reduced molecules M_1, M_2, \dots , it is sufficient to construct in addition the maps assembling vectors between each of these and every other and so obtain the synthetic vector maps of proposed structures of a whole series of crystals. Further, the method can be extended to deal with media at different levels.

To test a proposed structure of a crystal on the intensities we have to compare the synthetic vector map with the 'experimental' vector map, which is the Fourier transform of these intensities. All the experimental vector maps of protein crystals which are recorded in the literature are calculated from relative intensities and thus correspond to distributions in S on an unknown scale. It is possible to make rough estimates of the number of electrons in the unit cell. However, the scale on which the square of this number is to be inserted at the origin is unknown. Thus in all these calculations $I(000)$ is perforce taken to be zero. It results that all such experimental vector maps are reduced vector maps with total weight $w_v = 0$ and the distributions in S about which they afford information are reduced crystals, in which an electron density $g(x, y, z)$ —on an unknown scale—is replaced by $r(x, y, z) = g(x, y, z) - \bar{g}$ to yield a total weight $w = 0$. Hence these experimental vector maps relate to electron-density deviations in atomic space S , deviations from the average \bar{g} , not to

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electron densities, just as the synthetic vector maps in the reduced form in which they are more readily calculated refer to deviations from the level of medium in S . Since \bar{g} and m may differ, it is necessary to take account of a possible difference in level when comparing synthetic and experimental vector maps as also to bear in mind the scales of the two maps.

The importance of focussing attention on the crystal

entry in S_0 , vectors from every point in the unit cell make a contribution. No approximation to the vector map is obtained when the medium is neglected. Correspondingly, if it is claimed that some particular structure in a protein hydrate is detected from an experimental vector map, the claim can only be substantiated in terms of the proposed structure reduced to the level of the medium, not in terms of the proposed structure.

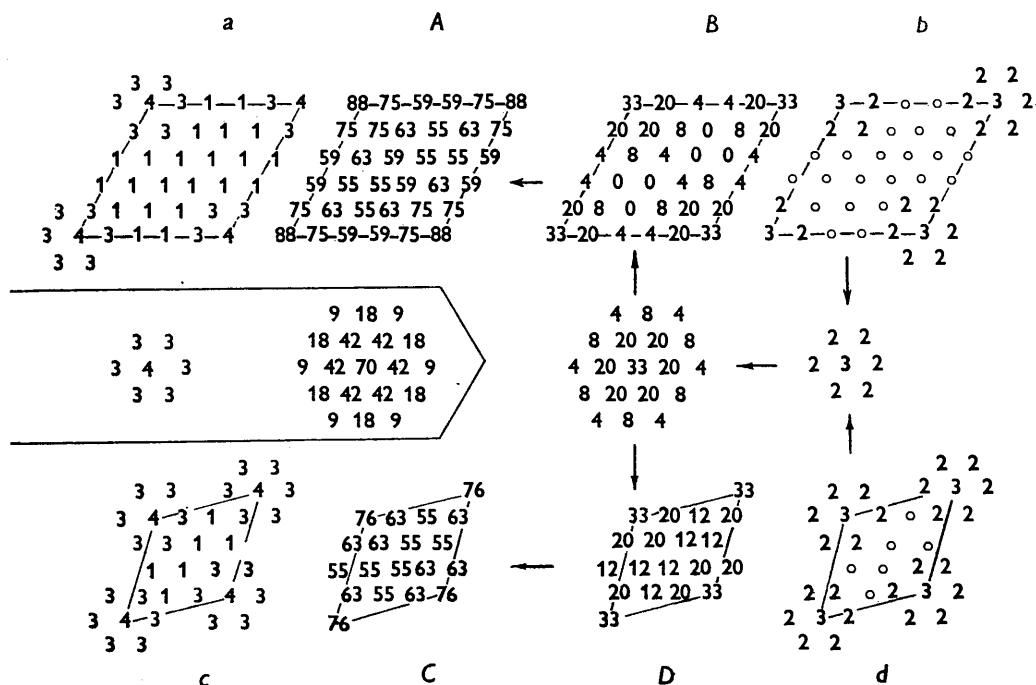


Fig. 1.

reduced to the level of the medium is apparent, when synthetic maps are under construction. However, this procedure has a crucial significance in the interpretation of experimental vector maps in general. For suppose a vector map such as B or D , but at the level corresponding to $w_0 = 0$, has been obtained by calculation from observed intensities, with $I(000)$ taken to be zero. How will the presence of the 'molecule' be detected? Comparing the vector map with the vector map of the molecule (shown between A and C) we see that not even in the neighborhood of the origin is the experimental map symptomatic of the molecule, however its level be adjusted. Such an experimental vector map is not the vector map of the crystal comprising only the molecule or molecules, since to every

There are two practical results of these findings: (1) In the study of the language of vector space, which would seem to be a necessary preliminary to the interpretation of experimental vector maps of proteins, the focus of interest is reduced structures represented by entries of both signs. (2) When the crucially important intensities observed in X-ray studies of proteins are recorded in the form of vector contour maps, the topology of the positive and negative regions forms an integrated whole. It may then be suggested that negative contours, as well as positive contours, be recorded in such maps.

Reference

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The unit cell and space group of ammonium metavanadate, NH_4VO_3 . By JOSEPH S. LUKESH, *Knolls Atomic Power Laboratory,* General Electric Company, Schenectady, N.Y., U.S.A.*

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The crystal chemistry of the vanadates is of interest because of the possibility of the existence of a series of

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framework structures analogous to those of the silicates. Ketelaar (1936) has shown that the oxide, V_2O_5 , is built of vanadium-oxygen tetrahedra which are linked by the sharing of three oxygen atoms of each tetrahedron by two vanadium atoms. This would be the analogue of SiO_2 where the tetrahedra share all four oxygen atoms. Serum